Synthesis of high-purity Ti2AlC by spark plasma sintering (SPS) of the elemental powders

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Recently, the layered ternary carbide $Ti₂AIC$ has received considerable attention since it is a remarkable material that combines many of the best attributes of both metals and ceramics [1]. Like metals it is an excellent electrical and thermal conductor, easily machinable, relatively soft, not susceptible to thermal shock and it behaves plastically at higher temperatures. Like ceramics it is oxidation resistant, refractory and has a high strength, high melting point and thermal stability.

 $Ti₂AIC$ was first synthesized and its structure was elucidated in the early 1960s [2]. In recent decades, various processes were employed to synthesize bulk Ti2AlC samples, however, it is very difficult to fabricate phase-pure Ti2AlC. More recently, Barsoum *et al*.[3, 4] successfully fabricated high-purity $Ti₂AIC$ polycrystal by reactively hot-pressing and high isostatic pressing (HIP) a mixture of Ti, graphite and Al_4C_3 powders. Wang *et al.* [5] also synthesized polycrystalline Ti₂AlC by using a solid-liquid reaction and simultaneous densification method. However, these processes are very complex. Thus further work is needed to develop new methods for the synthesis of $Ti₂AIC$. In the present research, spark plasma sintering of the elemental powders is reported. Elemental powders of Ti (99.0% pure, 10.6 μ m), Al (99.8% pure, 12.8 μ m) and carbon black (99%, 13.2 μ m) (all from the Institute of Non-Ferrous Metals, Beijing, China) were used to synthesize $Ti₂AIC$. The mixture was first mixed leave ethanol for 24 h. Then it was filled into graphite crucibles, prepressed at 20 MPa, and sintered in vacuum (0.5 Pa) in the temperature range $1100\degree$ C–1300 \degree C for 8 min by using a spark plasma sintering system (Mode SPS-1050, Lzumi Technology Co. Ltd). The heating rate was controlled at 80° C/min and the applied pressure was maintained at a constant 30 MPa during reaction sintering. The synthesized samples were 20 mm in diameter and 4–6 mm in thickness. The sintered product was characterized by X-ray diffraction (XRD) using a rotating anode X-ray diffractometer (Model D/MAX-RB, Rigaku Corporation, Japan). The microstructures of the samples were investigated via scanning electron micrographs (SEM) (Model JSM-5610LV, Jeol Ltd., Japan), coupled with energy-dispersive spectroscopy (EDS) for chemical analysis (Model Phoenix, EDAX, USA).

It has been reported [6, 7] that the purity of $Ti₃SiC₂$ synthesized from Ti/Si/C powders is very sensitive to departures from the stoichiometric composition. Like Si, Al evaporates easily when heated at high temperature. So a deficiency in Al favors the formation of TiC. In the present research, samples with three different starting molar ratios (a) Ti:Al: $C = 2:1:1$; (b) Ti:Al: $C =$ 2:1.1:1; (c) Ti:Al: $C = 2:1.2:1$ were investigated.

Fig. 1 shows the X-ray diffraction patterns of resultant products. The main phase in sample (a) was Ti2AlC, but it contained quite a large amount of TiC. No second phase was identified by X-ray diffraction in samples (b) and (c). This indicates that their products were pure $Ti₂AIC$. In sample (c), the X-ray diffraction peaks of $Ti₂AIC$ were much stronger than those of sample (b). The results confirmed the expectation that the excess of Al would favor the synthesis of $Ti₂AIC$. Therefore, the composition of sample (c) was chosen for use in the following experiments.

Fig. 2 shows the X-ray diffraction patterns of the resultant products sintered at different temperatures from 1100 °C to 1300 °C. At 1100 °C, no phase but Ti₂AlC wasidentified by X-ray diffraction, which indicated that the products were of high-purity $Ti₂AIC$. For the sample sintered at $1200\,^{\circ}\text{C}$, the peak of Ti₂AlC became weaker and a very strong peak of TiC was identified by X-ray diffraction. When the sintering temperature reached 1300 ◦C, TiC was also present, but the peaks were weaker than at $1200\degree C$ and the X-ray diffraction peaks of Ti2AlC almost disappeared at the same time. The main phase was $Ti₃AIC₂$, which indicated that at 1300 °C, Ti₂AlC reacted with TiC to form Ti₃AlC₂. This result is consistent with previous work [5].

The measured lattice parameters of $Ti₂AIC$ were $a =$ 0.3058 nm, $c = 1.3649$ nm; these are very close to these reported by other authors [4, 5]. The measured density of bulk material sintered at $1100\degree$ C was 4.10 g/cm³, which is 99.8% of the theoretical density of $Ti₂AIC$. It can be concluded that high purity $Ti₂AIC$ material can be synthesized by SPS from the starting powders mixtures with a molar ratio Ti:Al: $C = 2:1.2:1$.

Fig. 3 shows the scanning electron micrographs of the fracture faces of samples sintered at 1100 ◦C. In sample (a), two kinds of grains can be seen from the micrographs. Energy-dispersive spectroscopy (EDS) analysis revealed that the smaller grains in flocculent form were TiC. The larger laminated grains were $Ti₂AIC$. In sample (b) the $Ti₂AIC$ was tabular, which is quite analogous to the crystalline shape of $Ti₃SiC₂$. The average diameter of the grains was about 20 μ m, and there were about 5 μ m in thickness. The layered nature of the $Ti₂AIC$ grains is clearly seen in the fracture surface. No

Figure 1 XRD patterns of sample (a) with the starting composition of $Ti₂Al₁C$; (b) with the starting composition of $Ti₂Al_{1.1}C$; and (c) with the starting composition of $Ti₂Al_{1.2}C$.

Figure 2 XRD patterns of samples sintered at different temperatures (a) 1100 ◦C; (b) 1200 ◦C; and (c) 1300 ◦C.

other phases appeared. These results also show that a deficiency in Al usually favors the formation of TiC.

More importantly, the material had the same machinability as graphite. It could easily be machined with ordinary mechanical machining tools, and holes could readily be drilled by using common steel drills without adding lubrication.

It is concluded that high-purity polycrystalline bulk $Ti₂AIC$ was synthesized by spark plasma sintering of an elemental powder mixture with a molar ratio of Ti:Al: $C = 2:1.2:1$. The ideal synthesis temperature was

Figure 3 SEM photographs of the fracture faces of Ti₂AlC material synthesized from different starting compositions: (a) Ti₂Al₁C and (b) $Ti₂Al_{1.2}C.$

1100 ◦C, which is the lowest temperature for fabricating high-purity $Ti₂AIC$ material reported so far.

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